# Thickness Effect on Electronic Transport in Single Crystal La<sub>0.67</sub> Ca<sub>0.33</sub> MnO<sub>3</sub> Thin Films

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Abstract:  $La_{0.67}Ca_{0.33}MnO_3(LCMO)$  films were fabricated on (100) SrTiO\_3(STO) and (110) NdGaO\_3(NGO) substrates using facing-target sputtering technique. The resistivity and metal-to-semiconductor transition temperature  $T_{MS}$  were studied as functions of thickness. Although lattice stresses in LCMO/NGO and LCMO/STO are substantially different as confirmed by the XRD analyses, there is no direct correlation between structure and electronic transport in LCMO films, indicating the absence of strain effects on the transport properties, for example, ,  $T_{MS}$  and activation energy  $E_a$ .

Key words : colossal magnetoresistance ; thin film ; rare earthsCLC number : O484.1Document code : AArticle ID : 1002 - 0721(2005)06 - 0721 - 03

Lattice strain in thin films induced by film-substrate mismatch is one source for variant magnetic and resistive behaviors especially for the manganite perovskites for which the lattice effects are well established<sup>[1~5]</sup>. It has been also reported that lowfield colossal magnetoresistance (CMR) effect is due to strain-induced magnetic anisotropy<sup>[6,7]</sup>. Moreover, when the film gets thinner, surface effects become increasingly important<sup>[8]</sup>. Effects associated with surface morphology as well as structure deformation can be controlled by film thickness. Jin et al.<sup>[9]</sup> studied thickness effects on CMR in  $La_{0.67} Ca_{0.33} MnO_3$ , and Sun et al. <sup>[10]</sup> studied  $La_{0.67}Sr_{0.33}MnO_3$  films and wanted to verify the presence of a dead layer. In this paper, the thickness dependence of structure and transport behavior of La<sub>0.67</sub>Ca<sub>0.33</sub>MnO<sub>3</sub> (LCMO) films was studied, and the effect of lattice strain in CMR films explored.

### 1 Experimental

Epitaxial LCMO films with different thickness were grown on (100) SrTiO<sub>3</sub> (STO) and (110) Nd-GaO<sub>3</sub> (NGO) substrates using facing-target sputtering (FTS) technique<sup>[11-13]</sup>. A mixture of high purity Ar and O<sub>2</sub>, total pressure 10 Pa, was used as the discharge gases with the flow rates adjusted by two mass flow controllers. Prior to film deposition, the vacuum

chamber was pumped down to  $1 \times 10^{-4}$  Pa, and then the targets were presputtered for 15 min. Immediately after each deposition, the vacuum chamber was backfilled with 100 Pa oxygen gas. The deposited film was then cooled to room temperature with the substrate heater power cut off. To optimize the oxygen stoichiometry and crystallization, post-annealing process is necessary for the as-deposited films. Our samples were post-annealed at 900 for 240 min in O<sub>2</sub> gas. The film structure was examined by the /2 X-ray diffraction method<sup>[11~13]</sup>, and the resistance as a function of temperature was measured by standard four-probe technique.

## 2 **Results and Discussion**

XRD -2 scan confirmed the epitaxial growth and high crystal quality. Taking the diffraction peak of the substrate as an internal standard, the out-of-plane lattice parameters of LCMO layer can be determined accurately. Fig. 1 shows the variation of lattice constants with film thickness  $d_{\rm L}$  in LCMO/STO and LCMO/ NGO. Different crystal distortions are observed. In LCMO/STO, the lattice parameters are essentially unaffected until the thickness is smaller than ~ 50 nm below which a sudden in-plane expansion and out-ofplane contraction of lattice take place. In contrast, deformation of the unit cell in LCMO/NGO is small.

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From 30 to 200 nm, the variation of the out-of-plane lattice constant is  $\sim 0.0056$  nm for LCMO/STO and  $\sim 0.0013$  nm for LCMO/NGO, respectively.

For LCMO/NGO and LCMO/STO, except for the thinnest samples, all curves exhibit typical metal-to-semiconductor transition at  $T_{\rm MS}$  (metal-to-semiconductor transition temperature). The most triking features of present finding are the strong thickness dependent of  $T_{\rm MS}$  and the resistivity at a fixed temperature of 290 K as shown in the Fig. 2. Thickness effects are weak in films thicker than 50 nm. However,  $T_{\rm MS}$  depends critically on thickness in ultrathin films and drops steeply from ~ 200 to ~ 50 K when the thickness varies from ~ 20 to 5 nm. The sharp upturn of , consistent with the sudden drop of  $T_{\rm MS}$ , suggests a close relationship between these two quantities. Early

investigations of resistivity of La<sub>1-x</sub> Ca<sub>x</sub> MnO<sub>3</sub> for T >



Fig. 1 Out-of-plane lattice constants of LCMO/STO and LC-MO/NGO films as functions of film thickness (The solid lines are drawn as a guide to the eyes only. The irr set is the XRD patterns of LCMO (200 nm)/STO and LCMO (200 nm)/NGO.)



Fig. 2 Thickness dependence of resistivity at 290 K and metalto-semiconductor transition temperature ( $T_{\rm MS}$ ) of LC-MO/STO (solid points) and LCMO/NGO (open points) films (The solid lines are drawn as a guide to the eyes only. The inset is the temperature dependence of resistance of LCMO (200 nm)/STO and LCMO (200 nm)/ NGO)

 $T_{\rm MS}$  indicate that resistivity increases roughly exponentially with inverse temperature in semiconductor regime. The nature of conduction mechanism is ascribed to variable range hopping, or a nearest-neighbor hopping model based on small polaron. By fitting the semiconductor part (high temperature regime) of -T curve to small polaron model ( $=_0 T \exp(E_a/k_B T)$ , where  $_0$  is resistivity at 0 K,  $k_{\rm B}$  is Boltzmann constant), the activation energy for small polaron conduction ( $E_a$ ) is deduced from the slope of each curve. As shown in Fig. 3,  $E_a$  has a well-defined correlation with  $T_{\rm MS}$ , and increase monotonically from 0. 086 to  $\sim 0.125$  eV corresponding to a decrease of  $T_{\rm MS}$  from  $\sim 260$  to  $\sim 50$  K. In special,  $E_a$  is independent of the substrate.

From our results, although lattice stresses in LC-MO/NGO and LCMO/STO are substantially different as confirmed by XRD analyses, there is no direct correlation between structure and resistivity in our LCMO films, indicating the absence of strain effects on transport properties.



Fig. 3 Correlation between  $E_a$  and  $T_{MS}$  of LCMO/STO (solid points) and LCMO/NGO (open points) films (The solid line is as a guide to the eyes only)

#### 3 Conclusion

The resistivity and metal-to-semiconductor transition temperature  $T_{\rm MS}$  were studied as functions of thickness in LCMO/NGO and LCMO/STO films. Although lattice stresses in LCMO/NGO and LCMO/STO are substantially different as confirmed by XRD analyses, there is no direct correlation between structure and electronic transport in LCMO films, indicating the absence of strain effects on transport properties, for example, ,  $T_{\rm MS}$  and activation energy  $E_{\rm a}$ .

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# Preparation and Compressive Stress Effect of Polymer Matrix RE-Fe Giant Magnetostrictive Composite

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Abstract: Polymer matrix RE Fe giant magnetostrictive composite (GMPC) was prepared using bonding and magnetic field forming technique, and magnetostriction of samples was measured for different compressive stress. The experimental results show that there is certain compressive effect in GMPC. And the influence of compressive stress on magnetostriction of sample was investigated. It offers essential reference for application and device design of GMPC.

Key words: magnetostriction; polymer matrix; composites; GMPC; compressive stress effect; rare earths

(See J. Chin. RE. Soc. (in Chin.), 2005, 23(6): 727 for full text)

721 Thickness Effect on Electronic Transport in Single Crystal La<sub>0.67</sub>Ca<sub>0.33</sub>MnO<sub>3</sub> Thin Films

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Out-of-plane lattice constants of LCMO/STO and LCMO/NGO films as functions of film thickness (The solid lines are drawn as a guide to the eyes only. The inset is the XRD patterns of LCMO (200 nm)/STO and LCMO (200 nm)/NGO.)

724 In-Situ High Temperature X-Ray Diffraction Study of Structure and Phase Transformation of Nd-FePt Alloys

> Cheng Gang<sup>\*</sup>, Gu Zhengfei, Zhou Huaiying, Wang Zhongmin, Zou Ruiping, Yuan Songliu

- J. Rare Earths, 2005, 23: 724 ~ 726
- 727 Preparation and Mechanical Properties of Misch Metal Based Bulk Metallic Gasses

Wu Xiaofeng<sup>\*</sup>, Qiu Keqiang, Meng Likai, Zhao Wei, Suo Zhongyuan

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Nd content

0.381

0.380

0.379

0.3785

Lattice parameter winm

XRD patterns taken from the cross-section of as-cast rods with different diameters for  $La_{55}Al_{25}$   $Cu_{10}Ni_5Co_5$  and  $Mm_{55}Al_{25}Cu_{10}$   $Ni_5Co_5$  alloys

Lattice parameter of FCC

for  $Nd_x Fe_{60.5-x} Pt_{39.5}$  alloys vs. Nd content

732 Effect of Lu<sub>2</sub>O<sub>3</sub> on Charge/discharge Performances of Spherical Nickel Hydroxide at High Temperature

Ren Junxia , Wang Xiaojian , Li% = Yuzhan , Gao% = Xueping , Yan Jie $^{*}$ 



Cyclic voltammetric of regular Ni  $(OH)_2$  and that mixed with 3.5 %  $Lu_2O_3$ 

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