



Oscillatory superconducting transition temperature in $\text{YBa}_2\text{Cu}_4\text{O}_8/\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3/\text{YBa}_2\text{Cu}_4\text{O}_8$ heterostructures

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Abstract

$\text{YBa}_2\text{Cu}_4\text{O}_8/\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3/\text{YBa}_2\text{Cu}_4\text{O}_8$ (Y-124/LCMO/Y-124) heterostructures were prepared by facing-target sputtering technique. The oscillatory behavior of superconducting transition temperature (T_C) with the thickness of LCMO (d_L) has been observed. The strongest nonmonotonic information in T_C - d_L curves appears clearly when d_L is larger than the critical thickness d_L^{CR} . The metal-insulator transition temperature (T_{MI}) can only be detected at $d_L > d_L^{\text{CR}}$. For LCMO/Y-124/LCMO trilayer films, the T_{MI} has also shown an oscillation with decreasing spacer layer thickness. These kinds of symmetric dependence on the spacer layer in LCMO/Y-124 systems suggest strongly the interplay of ferromagnetic and superconducting couplings.

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1. Introduction

Since the advent of improved growth capabilities, the possibility to study proximity effects be-

tween layers with dissimilar intrinsic properties, such as superconductivity and ferromagnetism, has initiated much research activity [1]. For ferromagnetic (F) layers sandwiched between superconducting (S) layers it is expected that the superconducting transition temperature (T_C) decrease monotonically with increasing magnetic layer thickness. In fact interest in this topic increased considerably after Wong et al. [2] showed

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a nonmonotonic dependence of T_C as a function of the Fe layer thickness in V/Fe superlattice with fixed V thickness. Shortly afterwards, the possibility for oscillation of T_C as a function of the ferromagnetic layer thickness in S/F multilayers was demonstrated theoretically [3]. It was shown that for specific F layer thickness the Josephson coupling between two S layers can lead to a junction with an intrinsic phase difference $\Delta\phi = \pi$ which, in turn, exhibits a higher T_C .

Theoretical treatments of superconductor-ferromagnet-superconductor junctions have been made over the years. A critical thickness of a ferromagnetic barrier, above which there is no overlap of superconducting wave functions and hence zero Josephson current, is predicted by Kuplevakhskii and Fal'ko [4]. On the other hand, Buzdin et al. [5] reported an oscillation of the critical current of the junction with the ferromagnetic barrier thickness. Experiments on low T_C Pb–Fe–Pb junctions by Claeson [6] indicated a critical thickness of 0.5 nm. This small value may be due to efficient pair breaking by spin flip scattering at the Pb-Fe interfaces. Radovic et al. [3] have suggested that the unusual effects predicated, e.g., the oscillatory T_C with changing the thickness of ferromagnetic layer should be much more pronounced in S/F superlattices with high T_C and short coherence length superconductor. Since the discovery of the high T_C superconductivity, tremendous experimental results have been reported. Due to the structural similarity of the two classes of perovskite compounds, high T_C superconductor (HTS) and colossal magnetoresistance (CMR) ferromagnet, the unique combination of the two oxide layers is possible. Kasai et al. [7] reported the observation of supercurrents in trilayer junctions with a 500 nm thick $\text{La}_{0.7}\text{Ca}_{0.3}\text{MnO}_z$ ferromagnetic barrier. Recently, Holden et al. [8] investigated ellipsometric measurements of the far-infrared dielectric properties of $\text{YBa}_2\text{Cu}_3\text{O}_7/\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3$ superlattices. Their results provided clear evidence that the free-carrier response is strongly suppressed in these structures as compared to the one in pure $\text{YBa}_2\text{Cu}_3\text{O}_7$ and $\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3$ films. Pena et al. [9] found experimental evidence for coupling between superconducting layers through ferromagnetic spacer in

$\text{YBa}_2\text{Cu}_3\text{O}_7/\text{La}_{0.7}\text{Ca}_{0.3}\text{MnO}_3$ superlattices consistent with a long-range F/S proximity effect. In our previous work, we found that the T_C of $\text{YBa}_2\text{Cu}_4\text{O}_8$ layer was strongly suppressed by ferromagnetic $\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3$ layers due to the magnetic proximity effect in $\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3/\text{YBa}_2\text{Cu}_4\text{O}_8/\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3$ sandwiches [10].

In this paper we report a complex T_C oscillation phenomena for $\text{YBa}_2\text{Cu}_4\text{O}_8$ (Y-124)/ $\text{La}_{2/3}\text{Ca}_{1/3}\text{MnO}_3$ (LCMO)/Y-124 heterostructures. Here, we choose Y-124 only because it is much more stable than $\text{YBa}_2\text{Cu}_3\text{O}_7$ in different oxygen environment. The strong dependence of the T_C on the LCMO layer thickness is studied. We find a long-range proximity effect which yields coupling between superconducting Y-124 layers through the ferromagnetic spacer layer.

2. Experimental

All samples were fabricated on $10\text{ mm} \times 3\text{ mm}$ (001) SrTiO_3 (STO) single crystal substrates from pure LCMO and Y-124 targets under a range of conditions by facing-target sputtering technique [11,12]. The film thickness was controlled by sputtering time with the deposition rate being carefully calibrated (0.05 nm/s for LCMO layer, 0.075 nm/s for Y-124 layer). The as-deposited films exhibited a mirror-like surface and were adherent well on all substrates employed. From the z -profile, the roughness in z -direction is less than 5 nm [10]. The surface morphology reveals the symmetry of the underlying lattice surface and shows oriented, layered, square islands of about the same size [10]. The resistance as a function of temperature was measured by the standard four-probe technique with CIP (current in plane) geometry and the distance between voltage contacts was fixed at 6 mm. The magnetic moment of the samples as functions of temperature and applied magnetic field was measured by utilizing a vibrating sample magnetometer. During the measurements, a magnetic field was applied parallel to the film surface. A small nonhysteretic contribution from the STO substrate was eliminated by separately measuring its diamagnetic response.

3. Results and discussion

Fig. 1 shows the temperature dependence of the relative resistivity ($\rho/\rho_{100\text{K}}$) and its temperature coefficient ($d(\rho/\rho_{100\text{K}})/dT$) of Y-124 (75 nm)/LCMO (75 nm)/Y-124 (75 nm) film. The resistivity curve shows the metal-insulator transition of LCMO layer at high temperature, and the cross point at low temperature arises from the superconducting transition of Y-124 layers. The metal-to-insulator transition temperature, T_{MI} and T_{C} are defined as the peak points in the differential curve as shown in Fig. 1.

The hysteresis loops of Y-124 (75 nm)/LCMO (75 nm)/Y-124 (75 nm) film are shown in Fig. 2 with the magnetic field applied parallel to the layers. Magnetization loops at 120 K shows a saturation field of 1000 Oe, and resembles that of a single LCMO film [13]. The sample displays a characteristic superconducting like hysteresis loop below the T_{C} and the flattened slop probably results from a small stray field due to the magnetic layer when $H > 0$ Oe. Measurement of the zero field cooling (ZFC) magnetic moment as a function of temperature (see the inset) shows a clear ferromagnetic transition at 170 K. At lower temperature (below 70 K), the signature of the diamagnetic response of the Y-124 layer shows up, reducing the total magnetic moment of the sample, which becomes negative at lower temperatures.

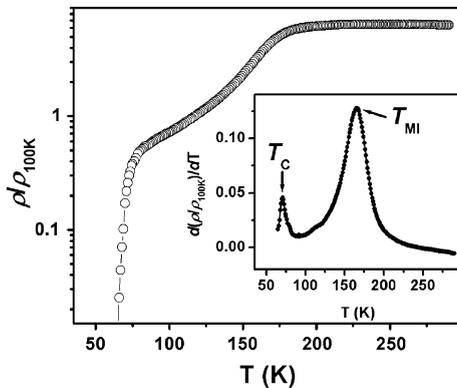


Fig. 1. The temperature dependence of the resistivity ($\rho/\rho_{100\text{K}}$) and its temperature coefficient ($d(\rho/\rho_{100\text{K}})/dT$) of Y-124 (75 nm)/LCMO (75 nm)/Y-124 (75 nm) film. T_{MI} and T_{C} are defined as the peak points in the differential curve as shown in the inset.

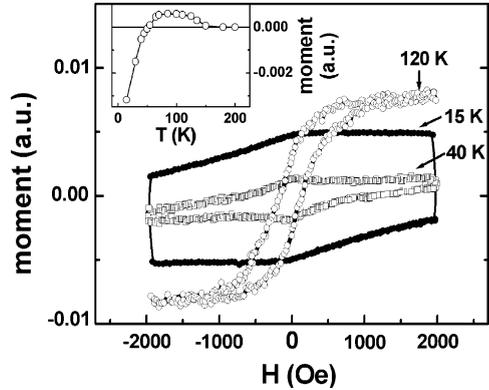


Fig. 2. Magnetic hysteresis loops of Y-124 (75 nm)/LCMO (75 nm)/Y-124 (75 nm) film with magnetic field parallel to the film plane at 15, 40 and 120 K, respectively. The inset displays the temperature dependence of ZFC magnetization with an applied field of 1000 Oe.

T_{MI} of as-deposited Y-124/LCMO/Y-124 sandwiches are plotted against the thickness of LCMO layer d_{L} in Fig. 3. T_{MI} is independent of its thickness for films with thick spacer layer. Degradation of T_{MI} has been found on films with thin LCMO layer. The extrapolation gives critical thickness d_{L}^{CR} below which the T_{MI} can not be detected. d_{L}^{CR} is about 5.5, 7.0 and 9.0 nm for $d_{\text{Y}} = 7.5, 45$ and 75 nm respectively, here, d_{Y} is the thickness of YBCO layer. Jin et al. studied the effects of film thickness for the $\text{La}_{0.67}\text{Ca}_{0.33}\text{MnO}_3$ film as early as 1995, and concluded that 100 nm is a promising thickness for CMR effect

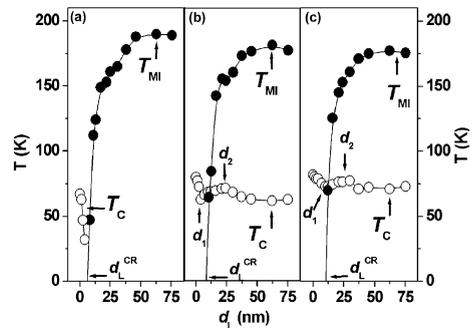


Fig. 3. T_{C} (○) and T_{MI} (●) of (a) Y-124 (7.5 nm)/LCMO (d_{L})/Y-124 (7.5 nm), (b) Y-124 (45 nm)/LCMO (d_{L})/Y-124 (45 nm) and (c) Y-124 (75 nm)/LCMO (d_{L})/Y-124 (75 nm) as functions of d_{L} . The lines are for a guide to the eyes.

[14]. Rao et al. accessed similar issue, and observed a significant enhancement of strain and an accompanied weakening of ferromagnetic order in ultra-thin film [15]. In a recent paper, Sun et al. found a sharp drop of T_{MI} in LCMO film if $d_L < 40$ nm, and even without MR transition for $d_L \leq 3$ nm [16]. In our case, the larger threshold value d_L^{CR} may be corresponding to the nonmagnetic (or magnetically ‘dead’) interface layer resulted from the roughness at the interface [17].

Fig. 3 also shows the d_L dependence of T_C for Y-124/LCMO/Y-124 trilayer films. For $d_Y = 7.5$ nm, T_C exhibits a rapid drop with increasing d_L and is not detected when $d_L > 6$ nm in our set-up. In the other two series, T_C exhibits a minimum with increasing d_L up to d_1 for the ultra-thin LCMO layer. As d_L is further increased from d_1 , T_C rises to a maximum at $d_2 \sim 30$ nm. The T_C is essentially independent of d_L for $d_L \gg 50$ nm and is ~ 65 and 70 K for $d_Y = 45$ and 75 nm when $d_L = 80$ nm. In a study of Fe/Nb system for different magnetic layer thickness, T_C reaches the first minimum d_{Fe} and ferromagnetism appears at almost the same Fe thickness as d_{Fe} [18,19]. In the Gd/Nb system the first minimum of T_C also correlates with the ferromagnetic ordering of the Gd layer [20,21]. In contrast, it should be pointed out that T_{MI} is smaller than Curie temperature T_f for our thin samples with low T_f and it is not sure that the magnetic proximity effect is still effective in the range, i.e., $d_L < d_L^{CR}$ since T_f of these samples can not be determined. So, d_1 cannot be directly connected to the onset of ferromagnetism in the LCMO layers.

The thickness dependence of resistivity at 90 K is shown in Fig. 4. It is interesting to note that the resistivity increases rapidly when d_L is increased from 0 to about 5 nm, a value close to d_1 above which the roughness at the interface becomes stable and independent of the increased layer thickness. d_1 may be partially resulted from the interface roughness effect. Although no evidence existed for nonmagnetic interfacial layers in Y-124/LCMO/Y-124 sandwiches, this mechanism, inelastic electron scattering in intermixed nonmagnetic layers at the interface may induce strong repulsive interaction on the Cooper pair, may play a role for the minimum of T_C at d_1 [18,19]. Fur-

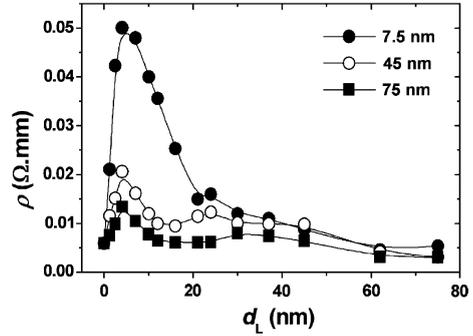


Fig. 4. Dependence of resistivity of Y-124/LCMO/Y-124 sandwiches at 90 K on spacer layer thickness d_L . The lines are for a guide to the eyes.

thermore, the maximum in T_C occurs at $d_2 > d_L^{CR}$. It is inferred that magnetic spin-flip scattering mechanism may be dominant with increase of d_L in an intermediate d_L region where ferromagnetic spin configuration is not so rigid due to two-dimensional character of magnetism of the LCMO layers, and play an important role in causing the maximum of T_C at d_2 [22].

T_C and T_{MI} of the as-deposited LCMO/Y-124/LCMO films are plotted against d_Y in Fig. 5. For films with thick spacer, T_C is independent of its thickness. Degradation of T_C has been found for films with ultrathin Y-124 layer. Extrapolation gives d_Y^{CR} below which the superconductivity is lost, and we have $d_Y^{CR} = 16, 35$ and 11.5 nm for

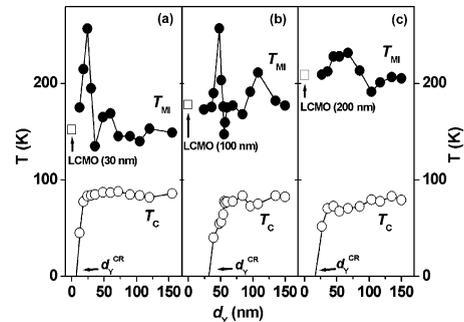


Fig. 5. T_C (○) and T_{MI} (●) of (a) LCMO (15 nm)/Y-124 (d_Y)/LCMO (15 nm), (b) LCMO (50 nm)/Y-124 (d_Y)/LCMO (50 nm) and (c) LCMO (100 nm)/Y-124 (d_Y)/LCMO (100 nm) as functions of d_Y . The lines are for a guide to the eyes. T_{MI} of LCMO single layer films (□) are shown in the figure.

the three series of $d_L = 100, 50$ and 15 nm, respectively. The results indicate that the superconducting properties are strongly affected by the ferromagnetic layers due to the magnetic proximity effect [10]. T_{MI} is close to the one of the individual LCMO layer for samples with thick Y-124 layer, and shows a nonmonotonic behaviour with decreasing spacer layer thickness. When $d_Y > d_Y^{CR}$ the peak values of T_{MI} are 257, 256 and 235 K for $d_L = 15, 50$ and 100 nm respectively, much higher than those for 30, 100 and 200 nm LCMO films, indicating that the LCMO layers are strongly coupled in those samples with thin YBCO spacer. The exchange coupling between the LCMO layers is also found in [LCMO (10 nm)/Pr_{0.67}Ca_{0.33}MnO₃(PCMO)]_n superlattices when PCMO thickness is less than 2 nm, where the T_{MI} is enhanced [23]. The striking discovery in T_{MI} of our work shows that the magnetic interaction between the two LCMO layers is still survived when $d_Y > 20$ nm.

The strongest nonmonotonic behaviors in T_C-d_L (Fig. 3) and $T_{MI}-d_Y$ (Fig. 5) curves appear clearly at the region $d_L > d_L^{CR}$ and $d_Y > d_Y^{CR}$. In the HTS/CMR structure, Y-124 has high critical temperature and short coherence length, while LCMO is metallic with not so large pair breaking effects due to its weak magnetism. The well-matched lattice between Y-124 and LCMO weakens the effects of roughness and strain. In addition, the oxygen diffusion between LCMO and Y-124 layers is negligible in our samples [10]. It is speculated that some coupling between the top and bottom layers, through the spacer layer, plays an important role.

A possible type of long-rang coupling is dipolar coupling due to the roughness of the interface [24–26]. The dipolar coupling between two perfectly flat infinite planes will vanish in the absence of roughness while in the presence of topographical inhomogeneities the magnetostatic coupling arising from surface magnetic dipoles created by roughness may originate a non-negligible coupling [26]. Also the dipolar interaction, under particular circumstances, can originate non-in-plane magnetizations. The roughness of samples is less than 5 nm [10]. We did not have the expression of dipolar interaction energy as a function of the distance

between them, but the dipolar coupling may play a role in the oscillation with the spacer thickness. Furthermore, although pinholes may be present in LCMO/YBCO/LCMO films when $d_Y < d_Y^{CR}$ T_{MI} is not enhanced and is the same as that for LCMO films, indicating that the magnetic short due to pinholes is ruled out.

Another possible mechanism may be related to the spin fluctuations in Y-124 and LCMO layers. The sandwiches can be considered to be a layered structure with Mn–O and Cu–O conduction planes which are stacked apart from each other. Chahara et al. think that spin fluctuations may be a key to understand the phenomena in HTS/CMR system [27]. In contrast, Holden et al. reported recently the long-range charge transfer from the HTS layer into the CMR layer for HTS/CMR superlattices from optical measurements, and concluded that spin diffusion (driven by the gradient in spin polarization between CMR and HTS layers) can not lead to a long-range spin polarization of the charge carriers deep inside the HTS layers [8].

In a word, our results provide evidence for interplay of ferromagnetic and superconducting couplings in HTS/CMR layered systems. To date, we do not have a complete microscopic understanding of these features yet. Further studies are required before one can distinguish between these equally fascinating possibilities. These results may be stimulating for the development of a theory of the HTS/CMR proximity effect.

4. Conclusion

The interlayer coupling of superconducting Y-124 thin films through ferromagnetic LCMO layer has been investigated from the transport properties of Y-124/LCMO/Y-124 sandwiches. With the increase of the LCMO layer thickness, the T_C exhibited an oscillation with a local minimum and a maximum. The strongest nonmonotonic information in T_C-d_L curves appears clearly at the region $d_L > d_L^{CR}$. In LCMO/Y-124/LCMO trilayer films, with decreasing spacer layer thickness, the T_{MI} also shows an oscillatory behavior. The kind of symmetric dependence on the spacer layer suggests strongly the interplay of ferromagnetic

and superconducting couplings in HTS/CMR layered systems.

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